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3,300,467
PRODUCTION OF CRYSTALLINE ISOTACTIC CIS-1,4-POLYPENTADIENE WITH A Ti(OR)₄-AIR₃ CAT-

1,4-POLYPENTADIENE TO ALYST Giulio Natta, Lido Porri, Gianfranco Stoppa, and Antonio Carbonaro, Milan, Italy, assignors to Montecatini Edison S.p.A., Milan, Italy Edison S.p.A., Milan, Italy Filed Feb. 11, 1963, Ser. No. 257,665 Claims priority; application Italy, Feb. 12, 1962, 2,683/62, 22,654/62 10 Claims. (Cl. 260—94.3)

The present invention relates to certain novel polymers of 1,3-pentadiene and to a simple and effective process for preparing these polymers.

In our copending U.S. application 193,528, filed May 9, 1962, there is described crystalline 1,3-pentadiene polymers having a cis-1,4 enchainment of the monomeric units and a syndyotactic configuration of the asymmetric carbon atoms. These polymers have an identity period, 20 along the direction of the main chain, of about 8.5 A. Moreover, in the crystalline state, the infrared spectrum of these polymers shows the presence of characteristic bands which are absent from the spectrum of the molten polymer or from the spectrum of the same polymer in 25 solution. The most intense of these bands are at 11.67 µ, 10.8 m, 10 m and 8.85 m.

We have now surprisingly obtained from 1,3-pentadiene certain new crystalline polymers having a prevail-ingly cis-1,4 enchainment as is the case with the above 30 mentioned polymers. In contrast to the polymers of our copending application however, the present novel polymers possess an isotactic configuration of the asymmetric carbon atoms.

FIGURES 1 and 2 are representations of the infrared 35 spectrum of polypentadienes possessing an isotatic configuration of the tertiary carbon atoms and a syndyotactic configuration of the tertiary carbon atoms respectively. In FIGURES 1 and 2 the absorbance values are shown on the ordinate, while the wave lengths in microns are 40 given as the upper scale on the abscissa and the frequency in cm.-1 is the lower scale thereon. In each of these two figures solid lines indicate the spectrum of the poly pentadiene in solution or in the molten state, whereas the dotted lines show the spectrum of the polymer in the crystalline state.

FIGURE 3 shows the X-ray spectrum, registered with a Geiger counter, of a polypentadiene of the present in-The relative intensities are shown on the ordinate and the diffraction angles (2θ) are plotted on the 50 abscissa.

It is therefore an object of the present invention to provide certain novel cis-1,4 polymers of 1,3-pentadiene possessing the isotactic structure.

Another object is to provide a process for preparing these polymers.

Further objects of the present invention and some advantages thereof will become hereinafter apparent.

The differences between the polymers of the present invention and known cis-1,4-polypentadiene are shown by the data obtained upon X-ray and infrared examina-tion of the polymers.

The fiber spectrum of the present polymers indicate the presence of an identity period of about 8.1 A, which value could only indicate a cis-1,4 isotactic structure; the identity period of the polymer possessing the syndyo-tactic structure (as reported in U.S. Serial No. 193,528), being about 8.5 A.

Other significant differences are observed in the fiber 70 spectrum of these two polymers, which differences relate not only to the position of the diffraction spots but also

to the state of the intensity diffracted onto the first two layers

In the spectrum of the syndyotactic polymer, the total intensity diffracted onto the 2nd layer is higher than that diffracted on the 1st layer; while in the spectrum of the instant novel polymers the opposite is observed. This is a further demonstration of the presence of the

cis-1,4 isotactic structure in these new polymers.

The X-ray spectrum, registered with a Geiger counter, of our new polymer as shown in FIGURE 3, wherein the relative intensity is plotted on the ordinate and the diffraction angles 2θ (CuK α) on the abscissa, presents two peaks corresponding to diffraction angles $2\theta=17^{\circ}$ and 26=18.8° and appears to be different from that of the corresponding syndyotactic polymer due to the different ratios between the intensities of these two peaks.

The infrared spectrum of our new polymer, in the molten state or in solution (FIGURE 1, continuous line) differs only slightly from the spectrum of the cis-1,4polypentadiene exhibiting a syndyotactic structure (FIG-URE 2, continuous line). Like the spectrum of the latter, it presents an intense band in the zone around 13.3 µ, which is characteristic of inner double bonds of the cis

Remarkable differences are, however, noted in the spectrum of the two polymers in the crystalline state. the spectrum of polymer containing the isotactic structhre (FIGURE 1, dotted line) there are present bands which are not present in the spectrum of the dissolved polymer or melted polymer, the most intense of which are at 11.88μ , 10.8μ , 9.95μ , 9.12μ , and 8.87μ . The position of these bands therefore is generally dif-

ferent from that of the bands exhibited by containing the syndyotactic structure (FIGURE 2, dotted

A particularly good criterion for differentiating the two polymers on the basis of their infrared spectrum in the crystalline state is the position of the band representing the inner double bonds of cis type. In the case of the syndyotactic polymer this band is at about 13.2,, while in case of the isotactic polymer the band is at about 13.4 s.

The crystalline 1,3-pentadiene polymers having the isotactic cis-1,4 structure of the present invention are obtained with catalysts prepared by reacting aluminum alkyls with titanium alkoxides.

These catalysts, when used with butadiene and diole-fins of the formula, CH=CH-CR=CH2 (wherein R is an alkyl) such as, e.g., isoprene, give polymers having a 1,2-enchainment as shown in Italian Patent No. 538,453. Therefore it could not be predicted that the same catalysts would, when employed with 1,3-pentadiene, give polymers having the cis-1,4 isotactic structure.

In preparing the catalyst according to the present invention any organic aluminum compound having the general formula:

> AL-R: R

wherein R2 and R2 may be the same or different alkyl, alkylaryl or cycloalkyl groups containing 1 to 15 carbon atoms and R1 can be a hydrogen atom or an alkyl, alkylaryl or cycloalkyl group containing 1 to 15 carbon atoms, may be used.

Some examples of suitable organoaluminum compounds include: dimethyl aluminum monohydride, diethyl alumi-num monohydride, diisopropyl aluminum monohydride, dibutyl aluminum monohydride, dihexyl aluminum monohydride, dioctyl aluminum monohydride, ethyl isobutyl aluminum monohydride, ethyl propyl aluminum monohydride, ethyl hexyl aluminum monohydride, aluminum tri-

ethyl, aluminum triisobutyl, aluminum tridodecyl, aluminum tricyclohexyl, aluminum diethylmonobenzyl, etc.

Mixtures of a dialkylaluminum monohydride with an aluminum trialkyl in any ratio can also be used.

Suitable titanium compounds comprise any compound of the formula, Ti(OR)4, wherein R is an alkyl, aryl, alkylaryl or cycloalkyl group, containing 1 to 10 carbon

Some examples of suitable titanium compounds in-clude: titanium tetra-n-butoxide, titanium tetraisopropoxide, titanium tetraphenolate, and titanium tetraethoxide.

The preparation of the polymerization catalyst is preferably carried out in an aliphatic or aromatic hydrocarbon solvent by simply mixing the two reactants together, prefcrably at room temperature.

The Al/Ti ratio can vary within very wide limits, e.g between about 1:1 and 100:1, preferably between 3:1

The temperature range suitable for the polymerization is very wide. Thus, the polymerization can be conducted 20 at temperatures between about -100° and +100° C., preferably between -50 and +50° C.

As the polymerization solvent or diluent, any aromatic

or aliphatic hydrocarbon can be used.

If desired, it is also possible to operate in absence of 25 extraneous solvents by using an excess of the liquid monomer as the solvent.

The cis-1,4-polypentadienes obtained with the catalysts prepared from aluminum alkyls and titanium tetra-alkoxides possess a melting point of 40-45° C., which is slightly higher than that of natural rubber. This property renders these polymers particularly suitable for the use in the field of elastomers.

By vulcanization of the cis-1,4-polypentadiene of the present invention, according to known methods, it is pos- 35 sible to obtain vulcanized products which are amorphous at room temperature in the unstretched state, but which are crystalline in the stretched state. These vulcanized products are characterized by a low initial modulus and a high tensile strength. Moreover, owing to the particular 40 configuration of the chains due to the cis-1,4 enchainment, these polymers exhibit a high rebound elasticity, comparable with that of natural rubber.

Cis-1,4-polypentadienes possessing the isotactic structure and obtained by the process of the present invention 45 usually have a rather high molecular weight, e.g., intrinsic viscosities [17] of about 6 to 7 (100 cc./g.) determined in toluene at 30° C.

The cis-1,4-polypentadienes containing the isotactic configuration are, however, easily degraded during process-ing in the mixer. Therefore the direct production of a high molecular weight polymer in the polymerization does not present a drawback. In this regard, cis-1,4-polypentadiene exhibits a behaviour similar to that of natural rubber and different from that of cis-1.4-poly- 55 butadiene which, as is known, does not undergo degradation in the mixer, and must therefore be directly obtained

with the desired molecular weight by polymerization.

The vulcanized products, obtained from cis-1,4-polypentadiene possessing an isotactic structure, also possess 60 a higher tear strength than the vulcanized products obtained from cis-1,4-polybutadiene.

The crude polymerization products obtained with cata-lytic system comprising AlR₁R₂R₃ and Ti(OR)₄, in gen-eral, have a cix-1,4 unit content of about 60-70% (determined by the analytical method described hereinbelow).

Polymers with a higher content of cis-1,4 units can be obtained by removing the macromolecules having a lower cis-1,4 content from the polymerization product. 70 This can be achieved, for example, by using the method described in our copending U.S. Serial No. 193,528, i.e., by dissolving the polymer in benzene or in another suitable solvent (e.g., ethylether, ethylacetate, etc.) and then reprecipitating it with methyl ethyl ketone. By follow- 75 21% of double bonds of trans type

ing this technique the less sterically less pure macromolecules, which are more soluble because of their lower tendency to crystallize, remain in solution.

By repeating the solution and reprecipitation operations two or three times it is possible to obtain polymers having a content of cis-1,4 units of about 85-90%. For most uses, however, such as wherein the polymers are used as elastomers, the purification of the product is not neces-

INFRARED ANALYSIS OF CIS-1,4-POLYPENTADIENE

The polymer is examined in a carbon disulphide solution. The trans type unsaturation is determined from the optical density of the band at 10.35μ , and the unsaturation of the vinyl type is determined from the optical density of the band at 11μ , while assuming that the absorption coefficients for the two bands are 10×10^4 x mols-1 x ml. and 12×104 x cm.-1 x mols -1 x ml. respectively, which are the average values reported in the literature (see A. L. McMurray and V. Thornton, Anal. Chem. 24, 318 (1952))

The optical density of the band at 10.35 is read with respect to a base line drawn between 10.08 µ and 10.55 µ. That of the band at 11 µ is read with respect to a base line

drawn between 10.8 µ and 11.2 µ.

The percentage of unsaturation of trans type is calculated according to the formula

$$C_1\% = \frac{D_{10.55} \times 6.8 \times 10^2}{S \times P}$$

that of vinyl type with the formula:

$$C_{*i_k}\% = \frac{D_{11} \times 5.7 \times 10^2}{S \times P}$$

S=thickness of the cell in cm.

P=mg. of polymer dissolved in 10 ml. of carbon disulphide.

The percent unsaturation of the cis-1,4 type is calculated by the formula:

$$C_{\text{eis}}\% = 100 - (C_{\text{t}}\% + C_{\text{vin}}\%)$$

The following examples are given to illustrate some specific embodiments of the present invention without limiting the scope thereof.

All parts and proportions are by weight unless otherwise indicated.

Example 1

A polymerization is carried out in a 100 cc. glass test be. In to this test tube, after the removal of air therefrom and the introduction of anhydrous N2, the following substances are introduced while operating without any contact with air:

The temperature is brought to 0° C. and 15 cc. of 1,3pentadiene (99% of trans isomer) are added to the homogenized mixture. The monomer is added 2 to 4 minutes after the formation of the catalyst. The whole mass is then kept at 0° C, for 18 hours and the polymerization is stopped by the addition of methanol. The polymer is coagulated by the addition of an excess of methanol and is then dried under vacuum.

3 g. of a solid crystalline polymer possessing the isotatic structure are obtained. This polymer upon infrared analysis is shown to consist of:

71% of double bonds of cis type 8% of double bonds of vinyl type 30

The polymer is dissolved in benzene, reprecipitated with methyl ethyl ketone and then subjected again to infrared analysis. This polymer is shown to contain:

78% double bonds of cis-type 17% double bonds of trans type 5% double bonds of vinyl type

After a further purification by dissolution in benzene and reprecipitation with methylethyl ketone, the polymer is shown to exhibit the following characteristics:

(a) Infrared analysis:

85% of double bonds of cis type 12% of double bonds of trans type 3% of double bonds of vinyl type

- (b) The polymer is shown to be crystalline by X-ray examination at room temperature. The X-ray spectrum thereof, registered with a Geiger counter, is reported in FIGURE 3;
- (c) Intrinsic viscosity, determined in toluene at 30° 20, -7.04 (100 cc./g.);
- (d) Melting temperature (under the polarizing microscope) = 43° C

Example 2

The polymerization is carried out as in Example 1, but 25

by operating at +20° C. instead of at 0° C.

After 5 hours, 4.2 g. of crystalline polymer are obtained which by infrared analysis is demonstrated to consist of:

66% of cis double bonds 25% of trans double bonds 9% of vinyl double bonds

After purification by means of two dissolutions and reprecipitations with methyl ethylketone as described in Example 1, the polymer exhibits a cis-1,4 unit content 35 of about 85%, possesses an isotactic structure, and is crystalline under X-ray examination.

Example 3

The polymerization is carried out in the manner of 40 Example 1, but using toluene instead of benzene and operating at -20° C. instead of 0° C. After 60 hours, 3 g. of crystalline polymer having a

cis-1,4 unit content of about 71% and the isotactic structure are obtained.

After purification by repeated dissolution and repre- 45 cipitation, a product having a cis-1,4 unit content of 86% is obtained.

Example 4

The polymerization is conducted as described in Ex- 50 ample 1, using the following reactants:

	and the
Anhydrous benzene	80
Titanium tetraisopropylate	0,6
Al(C2H5)'s was all all all all all all all all all a	
Pentadiene (95% trans)	15

After polymerization at 0° C. for 24 hours and coagulation with methanol, a product having a cis-1,4 unit content of 69% and a crystallinity derived from the isotactic structure (as shown by X-ray examination) is obtained. 60 The rate of conversion is 45%.

After purification by dissolution in benzene and reprecipitation with methyl ethyl ketone, a product containing 87% of units with cis-1,4 enchainment and having an intrinsic viscosity of 5.84 (100 cc./g.) (determined 65 in toluene at 30° C.) is obtained.

Example 5

In this polymerization run there is employed 40 cc. tanium tetraethoxide. The polymers produced in eac of anhydrous n-heptane, in which the following substances 70 case are practically identical with those of Example 1. are dissolved at 0° C, in the sequence:

		C.C.
Titanium tetra	a-n-butoxide	_ 0.3
Al(C2H5)3		0.9
Pentadiene (9.	5% of trans)	_ 6.0

The polymerization is carried out at 0° C. for 2 days. 3 g. of slightly crystalline polypentadiene, possessing the isotactic structure and having the following composition are obtained:

52% of cis-1,4 units 36% of trans units 12% of vinyl units

Example 6

This polymerization is carried out by using the penta-

diene as the solvent.

0.2 cc. of Al(C₂H₃)₅ and 0.005 cc. of titanium tetraisopropoxide are dissolved in 10 cc. of pentadiene (96% of trans isomer). After 2 days, the polymerization is stopped by adding methanol thereto. 0.3 g, of a dry polymer with a cis-1,4 unit content of 50% and possessing the isotactic structure are obtained.

Example 7

By operating as described in Example 1, a polymerization is carried out using the following ingredients:

	CC,
Toluene	30
Aluminum triisobutyl	0.6
Ti tetra n-butoxide	0.2
Pentadiene	4.0

The polymerization is carried out at 0° C. and is stopped after 40 hours by the addition of methanol.

1.9 g. of polypentadiene having a cis-1,4 unit content of 66% and the isotactic structure are obtained. After a purification step as shown in Example 1, a product containing 82% of cis-1,4 units is obtained.

Example 8

Using the pure cis isomer of pentadiene, the polymer-ization is carried out with the following ingredients:

	Cc.
Anhydrous benzene	. 25
Al(C ₂ H ₅) ₃	0.6
Ti tetra n-butoxide	0.2
Pentadiene (100% cis)	4.0

After 24 hours at 0° C., followed by coagulation in methanol and drying, 0.8 g. of polypentadiene, under in-frared analysis showing a cis-1,4 unit content of 78%, are obtained. After the parification or enriching step, the cis-1,4 content of the product increases to 87%.

Examples 9 to 13

The polymerizations are carried out as in Example I, but instead of aluminum triethyl, equimolecular amounts of each of the following aluminum alkyls are used: aluminum trihexyl, aluminum trimethyl, aluminum tridodecyl, aluminum tricyclohexyl and aluminum diethylmonobenzyl.

Polypentadiene having a prevailingly cis-1,4 enchainment (about 70% in the crude polymer and about 85% after purification) and the isotactic structure, which is crystalline at room temperature is obtained in each case.

Examples 14 to 16

The polymerization is carried out as in Example 1, but instead of titanium tetrabutoxide, equimolecular amounts of each of the following compounds are used: titanium tetrabenzoxide, titanium tetraphenoxide and titanium tetraethoxide. The polymers produced in each

Example 17

100 parts by weight of cis-1,4 polypentadiene exhibit-75 ing the isotactic structure containing 75% of cis-1,4 units,

7		8
which is obtained according to the preceding examples, are mixed in a roll mixer with the following ingredients:		After purification by means of two dissolutions and re- precipitations with methyl ethylketone as described in Ex-
Phenyl 3-naphthylamine		ample 1, the polymer shows a cis-1,4 unit content of about 83% and is demonstrated to be crystalline under
Laurinic acid 2	5	X-ray examination.
Zinc oxide		Example 20
Vulcafor HBS ¹ 1.8 Sulfasan R ² 2.2		The polymerization is carried out substantially as in Example 18, but by using toluene instead of benzene and
Cyclahexylbenzethiasyl sulphonamide, product of ICI. Morpheline disulphide, product of Monsante Chem. Co.	10	operating at -20° C. instead of 0° C. After 50 hours, 2.2 g. of crystalline polymer having
The mix thus obtained is vulcanized in a press at the temperature of 150° C. for 40 minutes. The vulcanized product possesses the following characteristics:		a cis-1,4 unit content of about 65% and the isotactic structure are obtained. After the purification, a product having a cis-1,4 unit
Tensile strengthkg./cm. ² 190 Elongation at breakpercent 1240		content of 84% is obtained.
Modulus at 300%kg./cm. ³ 13		Example 21
Example 18		Operating as described in Example 18, the following reactants are used:
A 100-cc. glass test tube provided with a side tube for		Anhydrous benzene 30
applying vacuum and introducing nitrogen is used as the polymerization reactor. Into this test tube, after the re- moval of air therefrom and introduction of anhydrous		Amygrotts betzete 300 Titanium tetra n-butoxide 0.2 $Al(C_2H_0)_2H$ 0.35
N ₂ , the following substances are introduced in the following order:		The catalyst is aged at 20° C. for 5 minutes and 7 cc. of pentadiene (95% of trans isomer) are added.
Cc.		After a polymerization at 0° C, for 24 hours and co-
Anhydrous benzene 80 Titanium tetraisopropoxide 0.3		agulation with methanol, a product having a cis-1,4 unit content of 64% and an intrinsic viscosity of 2.92 (100
Aluminum diethyl monohydride	90.0	cc./g,) (determined in toluene at 30° C.) is obtained. After purification by dissolution in benzene and repre-
pentadiene (98% of trans isomer, 2% of cis isomer) are		cipitation with methyl ethyl ketone, a product containing 82% of units with cis-1,4 enchainment and the isotactic
added to the homogenized mixture. The monomer is added 2 to 4 minutes after the formation of the catalyst.		structure, is obtained.
The whole mass is kept at 0° C. for 18 hours and the polymerization is then stopped by addition of methanol.		Example 22
The polymer is coagulated by addition of an excess of methanol and is dried under vacuum.		Operating as in Example 18, the following reactants are used:
2.8 g. of a solid crystalline polymer are thus obtained which under infrared analysis appears to consist of:	40	Anhydrous benzene
66% of double bonds of cis type	40	Tjtanium tetra-n-butoxide 0.1 Al(C ₂ H ₆) ₂ H 0.17
10% of double bonds of vinyl type 24% of double bonds of trans type		The catalyst is aged at 20° C. for 20 minutes and 10 cc. of pentadiene (95% of trans) are added.
The polymer is dissolved in benzene, reprecipitated with methyl ethyl ketone and subjected again to infrared analysis with the following results:		After polymerization at 20° C. for 30 hours, and co- agulation with methanol, 2.5 g. of polymer similar to that produced in Example 4, are obtained.
Percent		
Double bonds of cis type		Example 23
Double bonds of vinyl type 8	90	By operating according to the procedure described in Example 18, the following ingredients are used:
After a further purification by dissolution in benzene and reprecipitation with methylethyl ketone, the polymer		Anhydrous benzene
exhibits the following characteristics:		Aluminum triethyl
(a) Infrared analysis:	ha	Aluminum diethyl monohydride
84% of double bonds of cis type 11% of double bonds of trans type		Pentadiene (95% trans) 7.0
5% of double bonds of vinyl type (b) The polymer is shown to be crystalline under X-	60	The monomer is added 1-2 minutes after the formation of the catalyst.
ray examination at room temperature; (c) Intrinsic viscosity (determined in toluene at		After 25 hours of polymerization, at 20° C. and after congulation with methanol, a product having a cis-1,4 unit content of 65% is obtained with a conversion of
30° C.) [n]=-5 (100 cc./g.); (d) Melting temperature (under the polarizing micro-		60%. After purification by dissolution in benzene and
scope) == 40° C. Example 19	65	reprecipitation with dimethyl ethylketone, a product con- taining the isotactic structure and 85% of cis-1,4 units is obtained.
The polymerization is carried out as in Example 18,		Example 24
but by operating at +20° C. instead of at 0° C. After 5 hours, 3.2 g. of a solid crystalline polymer possessing the isotactic structure are obtained which polymer under infrared analysis is shown to consist of:	70	Operating as described in Example 18, the following substances are used: Cc.
61% of cis double bonds		Anhydrous benzene80
276 of trans double hands		Al disobutylmonohydride 0.6 Ti tetra n-butoxide 0.2
12% of vinyl double bonds	75	Pentadiene (98% trans)
30 V. 11 V. 12 V.		/ / / /

The polymerization is carried out at 20° C., and is stopped by adding methanol to the reaction after 30 hours. 2.8 g. of polypentadiene with a cis-1,4 unit content of 64% are obtained. After the purification or enriching steps, the cis-1,4 content increases to 87%. The intrinsic viscosity of the crude polymer is 2.4 (100 cc./g.) (determined in toluene at 30° C.).

Example 25

Using the pure cis isomer of pentadiene, the polymerization is carried out using the proportions and conditions described in Example 21.

After polymerization at 0° C. for 24 hours, 1.2 g. of polymer, which by infrared examination is shown to have a cis-1,4 unit content of 76% (85% after purification) 15 are obtained.

Examples 26 to 28

The polymerization is carried out as in Example 18, but instead of titanium tetraisopropoxide, equimolecular amounts of each of the following compounds are used: 20 titanium tetrabenzoxide, titanium tetraphenoxide and titanium tetraethoxide. The polymers obtained are in each case practically identical with those of Example 18,

Many variations and modifications can, of course, be practised without departing from the scope and spirit of 25 the present invention.

Having thus described the present invention, what it is desired to secure and claim by Letters Patent is:

1. A process for preparing cis-1,4 unsubstituted polypentadienes in which essentially all the cis-1,4 units have 30 an isotactic configuration and which show at the X-rays crystallinity due exclusively to isotactic structure which comprises homopolymerizing 1,3-pentadiene in the presence of a catalyst obtained from (a) a titanium compound having the general formula:

wherein R represents a member selected from the group consisting of alkyl, aryl, alkylaryl and cycloalkyl groups and (b) an organo aluminum compound having the 40

wherein R2 and R3 each represents a member selected from the group consisting of alkyl, alkylaryl and cycloalkyl groups and R; represents a member selected from the group consisting of hydrogen atoms, alkyl, alkylaryl and cycloalkyl groups.

2. The process of claim 1, wherein titanium compound is selected from the group consisting of titanium tetran-butoxide, titanium tetraisopropoxide, titanium tetraphenolate and titanium tetraethoxide.

3. The process of claim 1, wherein the organic alu-minum compound is selected from the group consisting of dimethyl aluminum monohydride, diethyl aluminum

monohydride, diisopropyl aluminum monohydride, dibutyl aluminum monohydride, dihexyl aluminum monohydride, dioctyl aluminum monohydride, ethyl-isobutyl aluminum monohydride, ethyl-propyl aluminum monohydride, ethylhexyl aluminum monohydride, aluminum triethyl, trihexyl aluminum, triisobutyl aluminum, tridodecyl aluminum, tricyclohexyl aluminum and diethyl monobenzyl alumi-

4. The process of claim 1, wherein the molar ratio of organic aluminum compound to titanium compound is between about 1 and 100

The process of claim 1, wherein the molar ratio of organic aluminum compound to titanium compound is between about 3 and 15

6. The process of claim 1, wherein the polymerization is carried out at a temperature between about

7. The process of claim 1, wherein the polymerization is carried out at a temperature between about +50° C.

8. The process of claim 1, wherein the polymerization

is carried out in the presence of a hydrocarbon solvent.

9. The process of claim 1, wherein the polymerization is carried out using liquid 1,3-pentadiene monomer as the solvent.

10. The process for enriching the cis-1,4 unit content of an unsubstituted polypentadiene obtained by the process of claim 1, which comprises treating the polypentadiene with a solvent to thereby separate the sterically less pure macromolecules therefrom due to the higher solubility of said macromolecules.

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JOSEPH L. SCHOFER, Primary Examiner.

E. J. SMITH, Assistant Examiner.

Jan. 24, 1967

PRODUCTION OF CRYSTALLINE ISOTACTIC CIS-1.4-POLYPENTADIENE

WITH A Ti(OR)4-ARR3 CATALYST

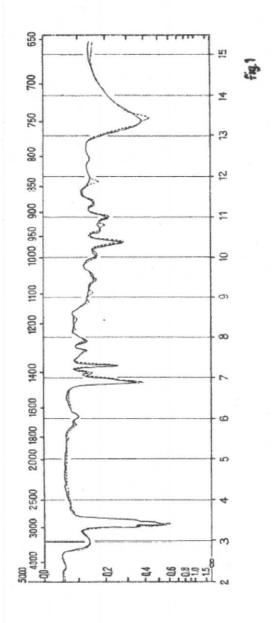
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WITH A Ti(OR)4-ARR3 CATALYST

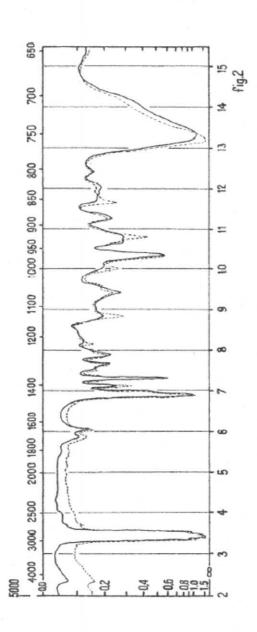
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Jan. 24, 1967

PRODUCTION OF CHYSTALLINE ISOTACTIC CIS-1,4-POLYPENTADIENE
WITH A TI(OR)4-AKR3 CATALYST
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WITH A TI(OR)4-AKR3 CATALYST
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PRODUCTION OF CRYSTALLINE ISOTACTIC CIS-1,4-POLYPENTADIENE

WITH A Ti (OR)4-ALR3 CATALYST

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Filed Feb. 11, 1963

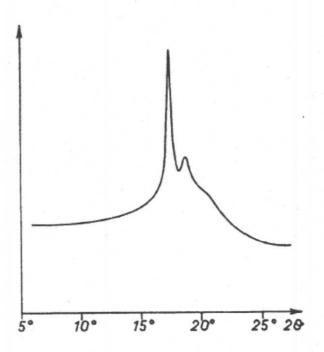


Fig. 3